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**Computing Grand Challenge Symposium**

**Toward Petascale Atomistic Simulations with  
Quantum-Level Accuracy**



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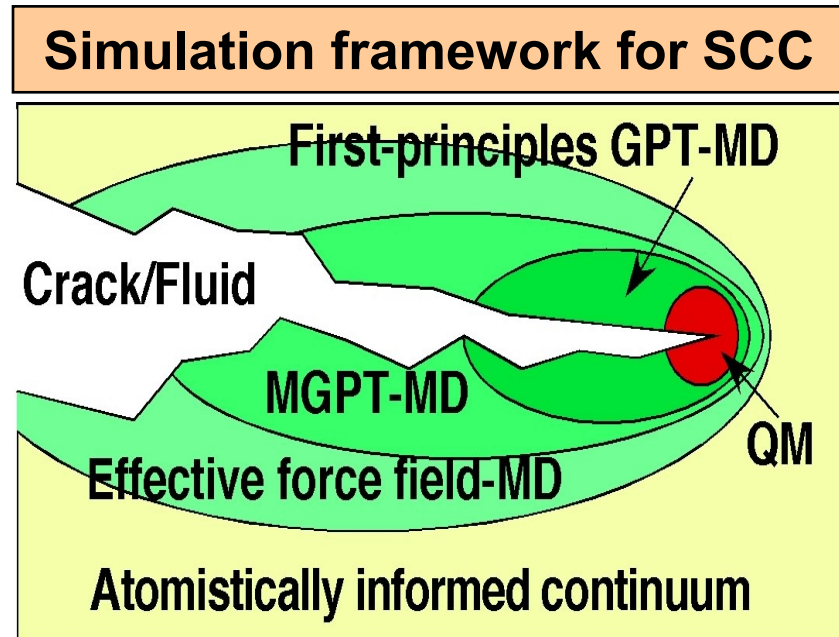
**Collaborations: USC (*Vashishta et al.*), Harvard (*Kaxiras et al.*),  
LANL (*Voter et al.*)**

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Lawrence Livermore National Laboratory under contract No. DE-AC52-07NA27344

# Motivation: recently funded SciDAC-2 project on stress corrosion cracking (SCC)



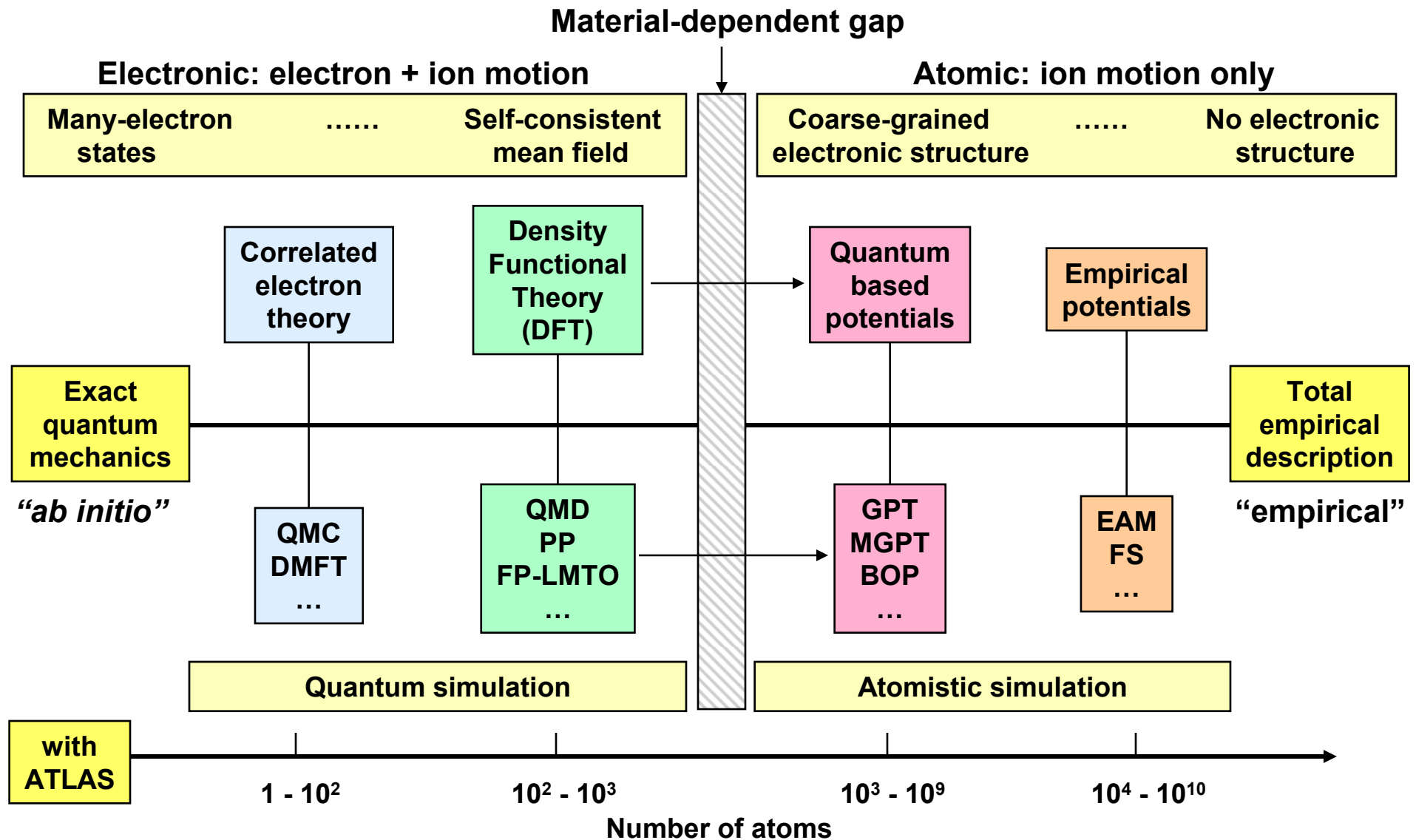
- Corrosion is complex technical problem (in e.g., advanced power generation) with annual economic impact equal to 3% of GNP
- USC, Harvard, CSUN, Purdue, LANL, LLNL collaboration
- **5-year goal**: develop hierarchical petascale simulation framework to address SCC from first principles
- **LLNL role**: develop new capability to perform ultra-scale atomistic simulations ( $10^6$ – $10^{12}$  atoms) with quantum-mechanical accuracy



Materials: *d*-electron transition metals and alloys, e.g., NiAl, at high temperatures

**SciDAC focus: advanced simulation capability development**

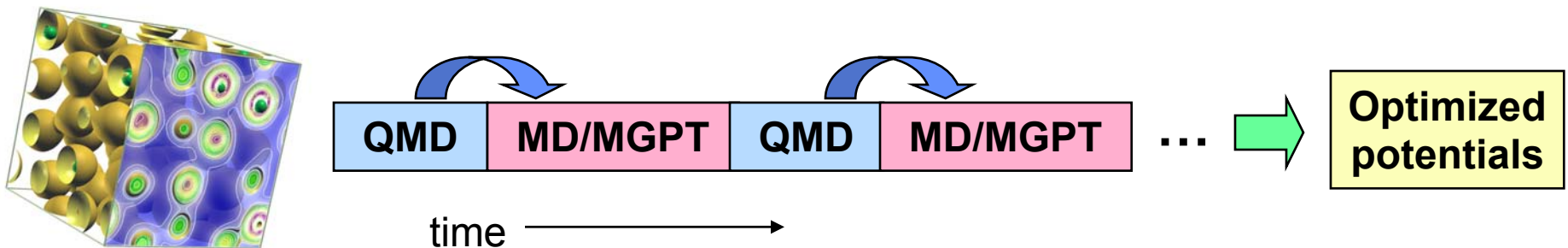
# Bridging the gap from quantum mechanics to large scale atomistic simulation



# Scientific challenge and strategy



- **Scientific challenge**: bridge the length and time scale gap from quantum mechanics ( $\sim 100$  atoms) to ultra-scale atomistic simulation (billions of atoms) with quantum-level accuracy
- **Strategy**: Direct coupling of *quantum simulations* (**QMD**) and MD simulations with quantum-based multi-ion potentials from *model generalized pseudopotential theory* (**MGPT**)
  - ATLAS enabling development of a robust **hybrid QMD-MD/MGPT simulation capability** for complex *d*-electron metals and alloys



- Grand challenge goal: simulate  $10^9$  atoms at high temperature in prototype systems (e.g., Mo, NiAl) via MD/MGPT with accuracy of direct QMD simulation on  $10^2$  atoms

## Scope of work and progress to date

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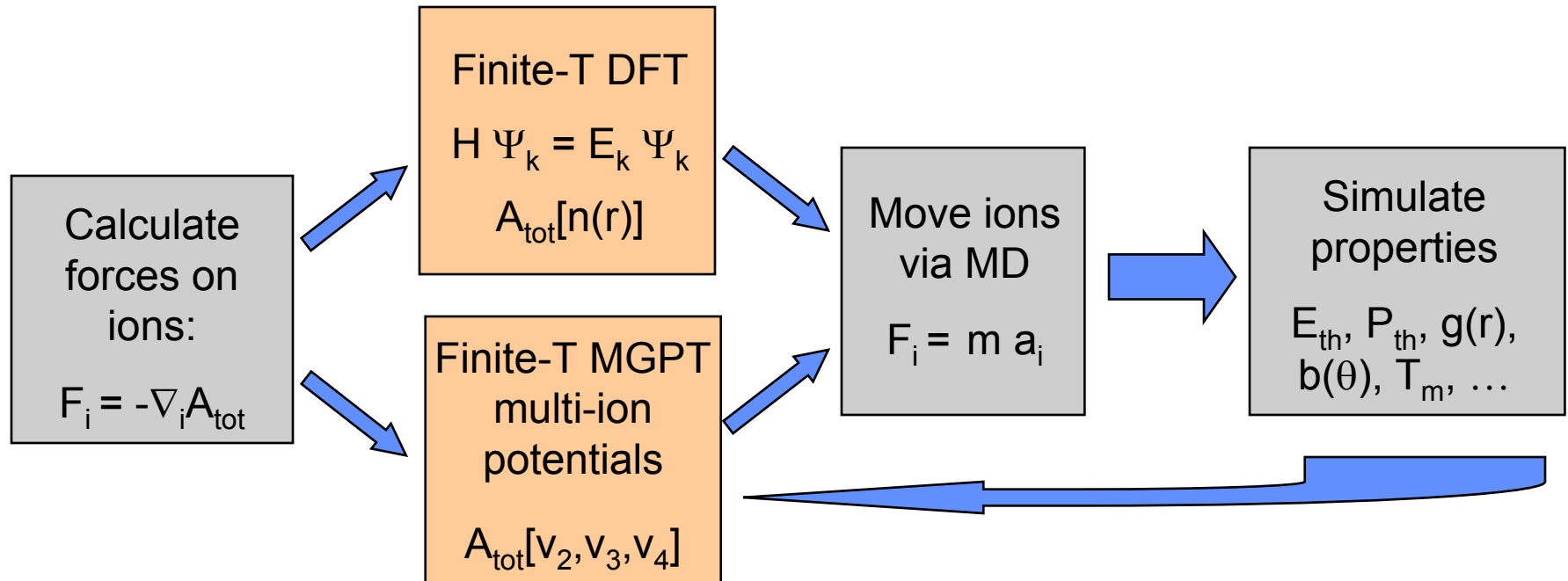


- 1) Development and testing of an efficient parallel QMD-MD/MGPT simulation code: *P<sup>3</sup>MD* **completed**
- 2) Development and application of robust simulation algorithm to optimize MGPT potentials
  - first-generation algorithm established and applied to Mo prototype
  - excellent set of potentials obtained that accurately predict atomic structure of liquid Mo **moving toward optimization**
- 3) Establishment of a QMD test data base and corresponding test simulations on derived MGPT potentials: *transferability*
  - extensive two-phase Mo melt simulations **in progress**
- 4) Billion-atom MD/MGPT demonstration simulations with optimized potentials **still to be completed**

# Hybrid QMD-MD/MGPT simulation code: *P<sup>3</sup>MD*



*P<sup>3</sup>MD* = Petascale, Plane-wave, Pseudopotential Molecular Dynamics



- **QMD mode: *P<sup>3</sup>MD* implements first-principles pseudopotential method**
  - plane wave basis for wavefunction expansion
  - treats 3-9 valence electrons per atom for *d*-electron transition metals
  - treats 50-250 transition-metal atoms with 1-10 k points
- **MD/MGPT mode: *P<sup>3</sup>MD* implements model generalized pseudopotential theory**

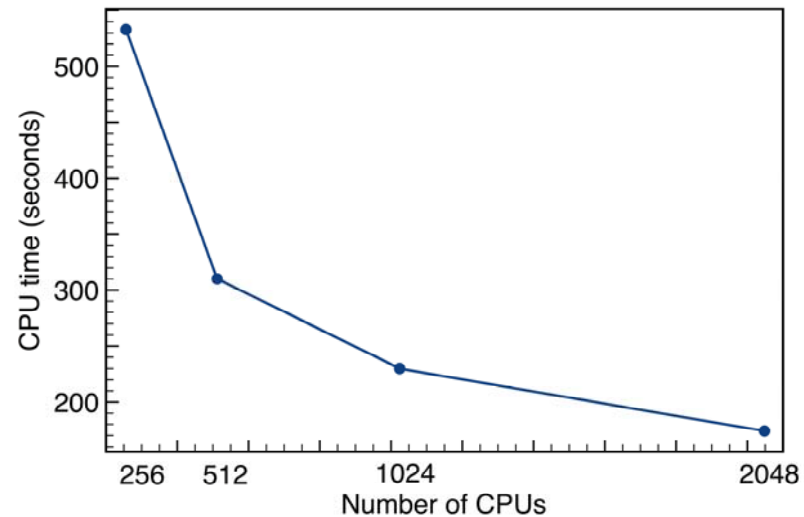
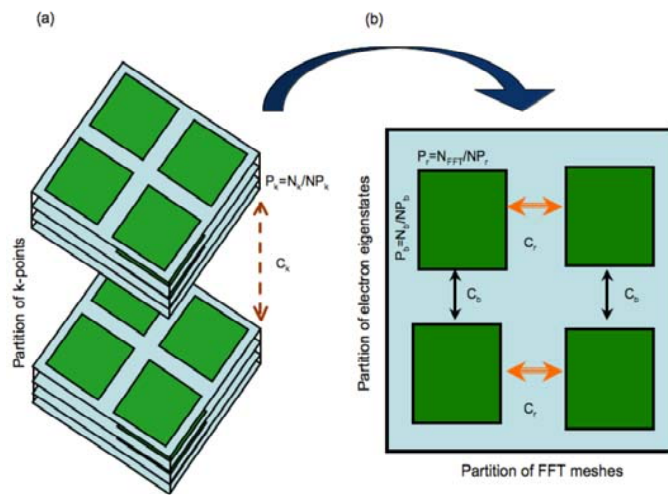
# Efficient usage of HPC resources



## Parallelization of $P^3MD$ code:

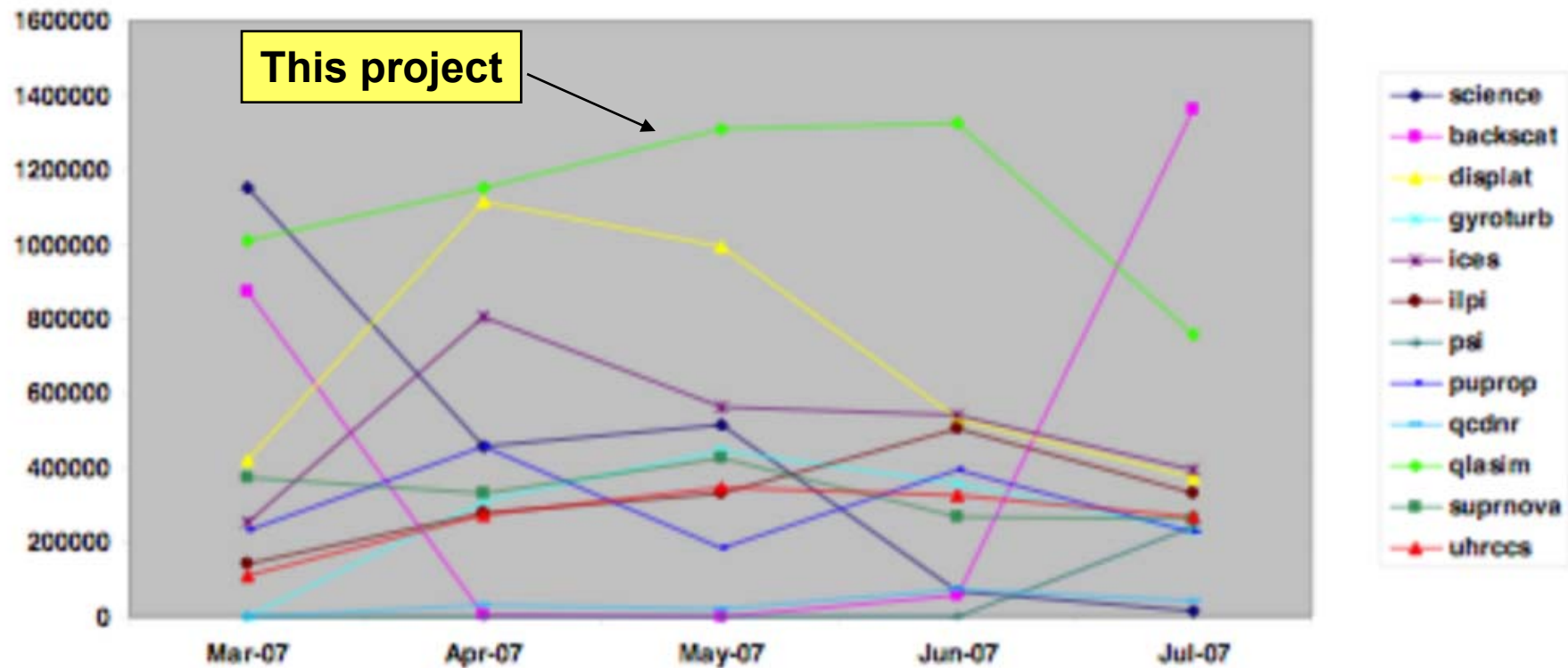
DFT forces:  $k$  points, energy bands, plane waves in reciprocal space spread across CPUs in optimal manner, with custom FFT used to move between real and reciprocal space

MD: performed on single CPU, assuming  $N$  is small ( $< 250$ )



**Tests on QMD melt simulations show excellent efficiency**

# Atlas utilization on grand challenge projects



**Average utilization by this project: 300,000 CPU/hours per week**



# Quantum-based MGPT multi-ion potentials



## Total-energy expansion within DFT quantum mechanics

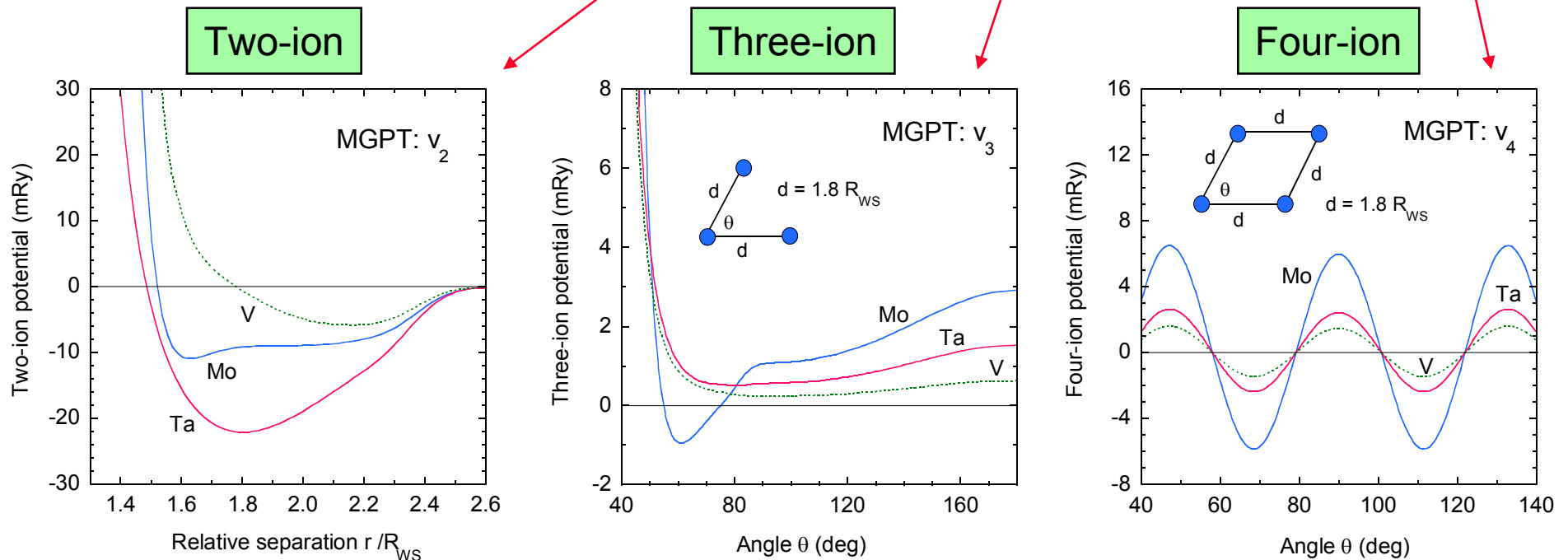
- MGPT potentials *derived* from systematic simplification of first-principles results
- standard analytic method: *canonical d bands* (universal form)
- fast matrix method: *non-canonical d bands* (materials specific form)

$$E_{tot}(R_1, \dots, R_N) = NE_{vol}(\Omega) + \frac{1}{2} \sum_{i,j} v_2(ij; \Omega) + \frac{1}{6} \sum_{i,j,k} v_3(ijk; \Omega) + \frac{1}{24} \sum_{i,j,k,l} v_4(ijkl; \Omega)$$

volume

radial forces

angular forces



# Importance of electron temperature $T_{el}$ : three limits

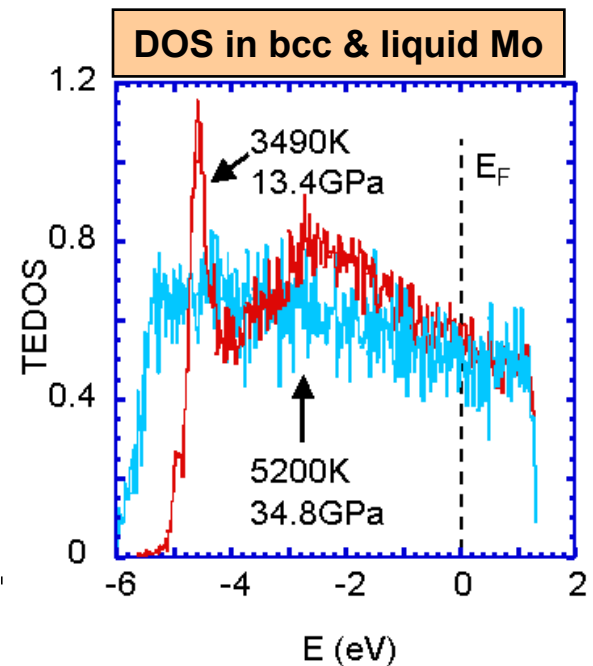


(i)  $T_{el} = 0$  and  $T = T_{ion}$ : **low-temperature solid**

- standard assumption in theory, even for high  $T$
- adequate for most simple  $sp$ -bonded materials

(ii)  $T_{el} = T_{ion} = T$ : **high-temperature solid and liquid**

- important in  $d$ -electron transition metals due to high, phase-dependent density of electronic states (DOS)
- also advantageous: allows one to leverage rich finite- $T$  QMD data
- in MGPT, include through explicit  $T$ -dependent potentials:



$$A_{tot}(R_1, \dots, R_N) = \underbrace{NA_{vol}(\Omega, T)}_{\text{volume}} + \underbrace{\frac{1}{2} \sum_{i,j} v_2(ij; \Omega, T)}_{\text{radial forces}} + \underbrace{\frac{1}{6} \sum_{i,j,k} v_3(ijk; \Omega, T)}_{\text{angular forces}} + \frac{1}{24} \sum_{i,j,k,l} v_4(ijkl; \Omega, T)$$

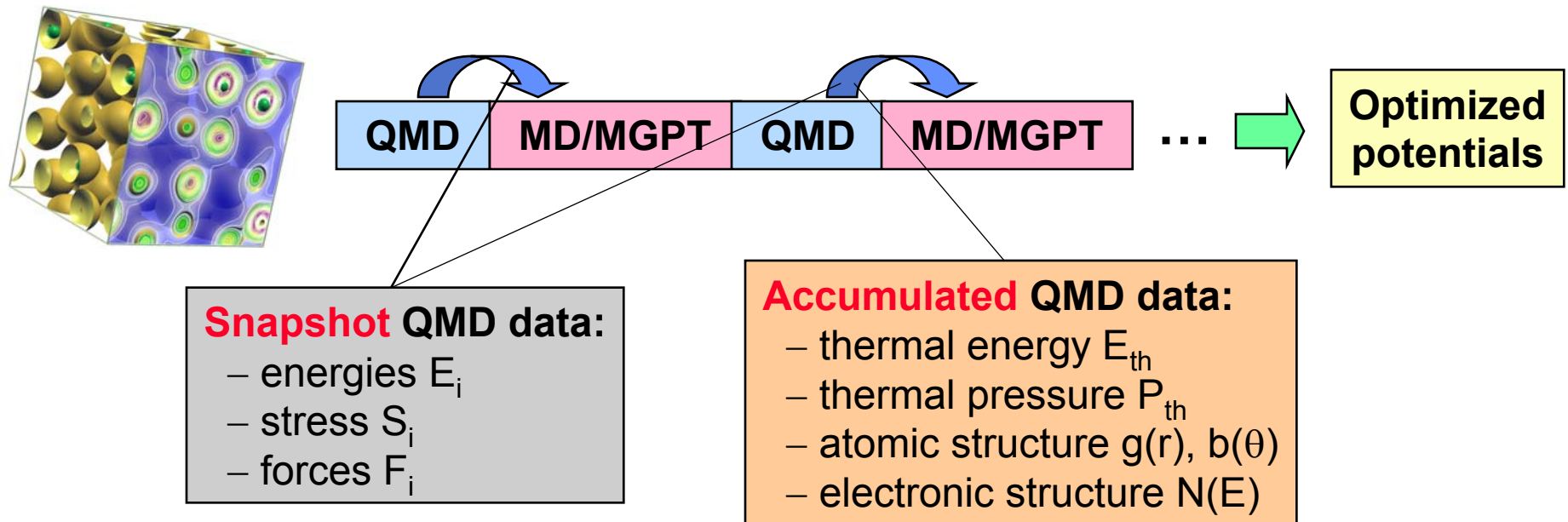
volume

radial forces

angular forces

(iii)  $T_{el} \gg T_{ion}$ : **ultra fast laser heating of solid or liquid**

# QMD-MD/MGPT simulation strategy



## Optimization considerations:

- details of MGPT potentials functional form
  - *canonical* vs *non-canonical*  $d$  bands
  - finite-T systematic improvements
- choice of constraining data
  - snapshot vs accumulated QMD data
  - additional static solid-phase data

**Various options have been tried and a first-generation simulation scheme adopted**

# Adopted first-generation simulation scheme



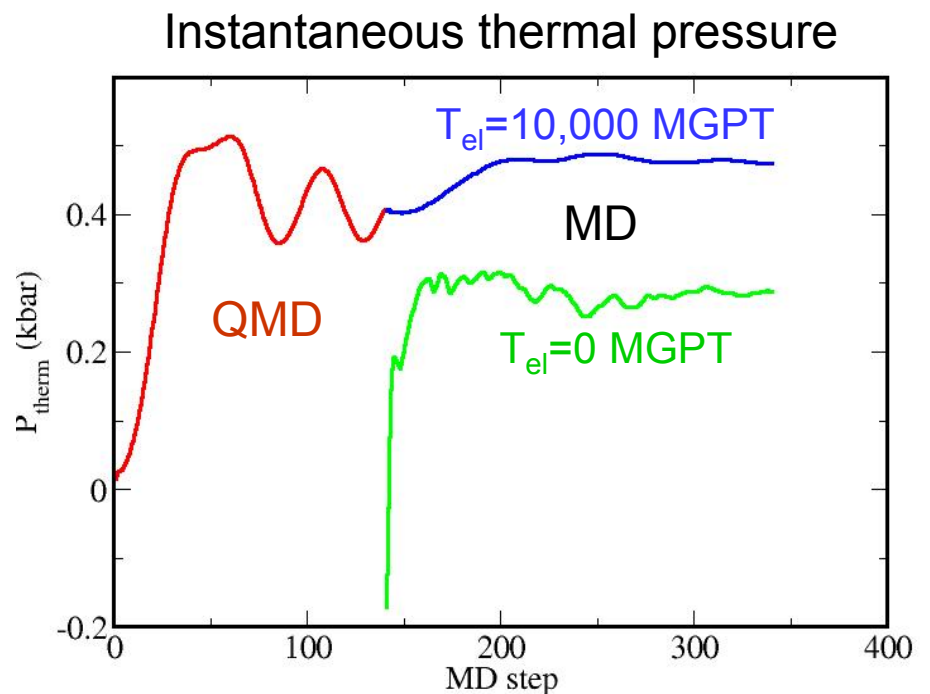
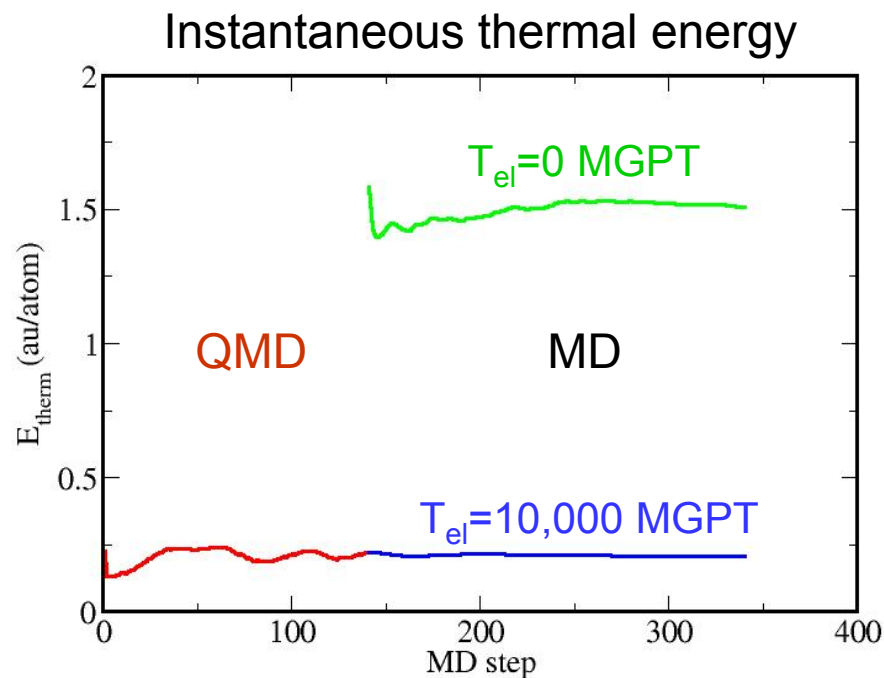
## MGPT potentials:

- *non-canonical*  $d$  bands
- explicit modeling of  $T_{\text{el}}$  contributions to  $A_{\text{el}}$

## Constraining data:

- blend of static bcc data at  $T_{\text{el}}$  + *snapshot* QMD energies in liquid from 20-30 configurations
- *predict* accumulated QMD properties as test

## Smooth QMD → MD/MGPT transition and rapid convergence:

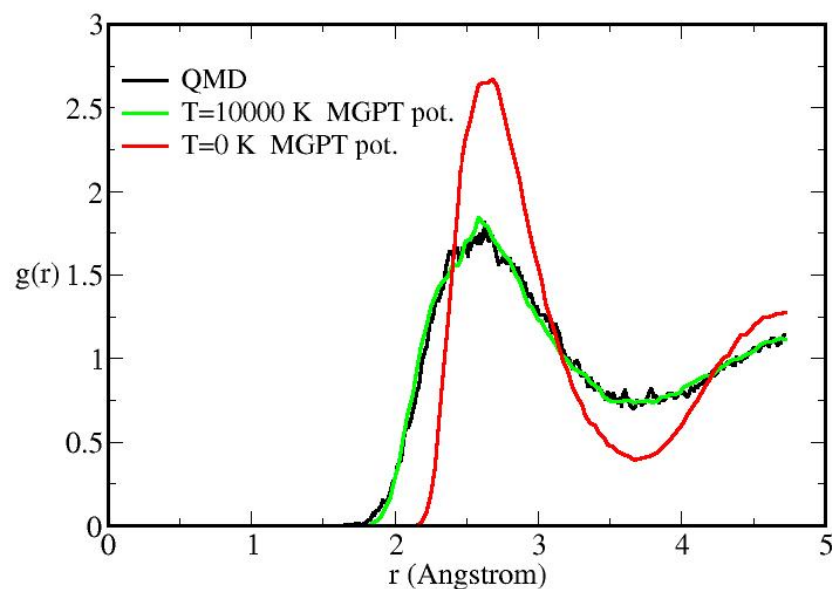


Simulation conditions:  $N = 54$  atoms,  $T_{\text{ion}} = 10,000$  K,  $\Omega = 105.1$  au

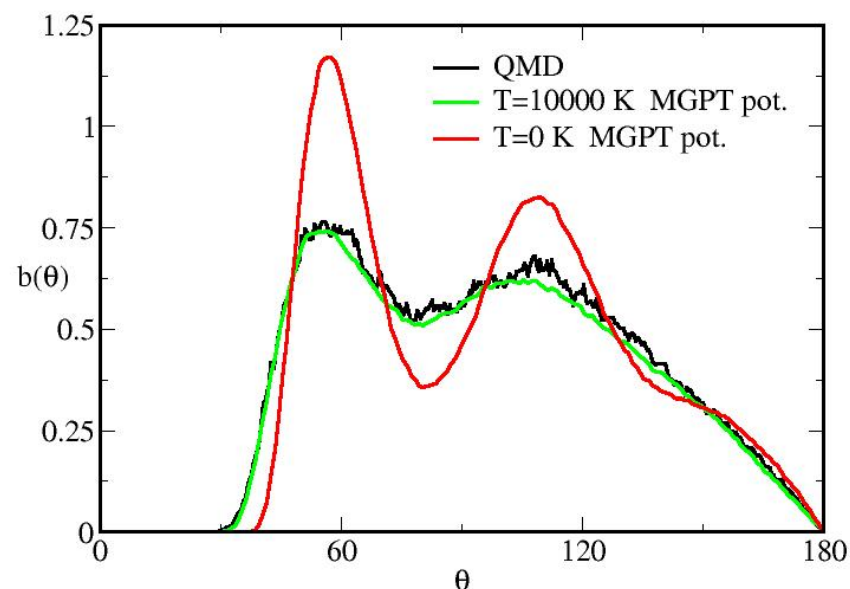
# First-generation Mo potentials accurately predict atomic structure of liquid



Radial distribution function  $g(r)$



Bond-angle distribution function  $b(\theta)$



Simulation conditions:  $N = 54$  atoms,  $T_{\text{ion}} = 10,000$  K,  $\Omega = 105.1$  au

**Similar results have been obtained over a wide range of volumes at both 5000 and 10,000 K**

# Two-phase baseline QMD melt simulation in Mo

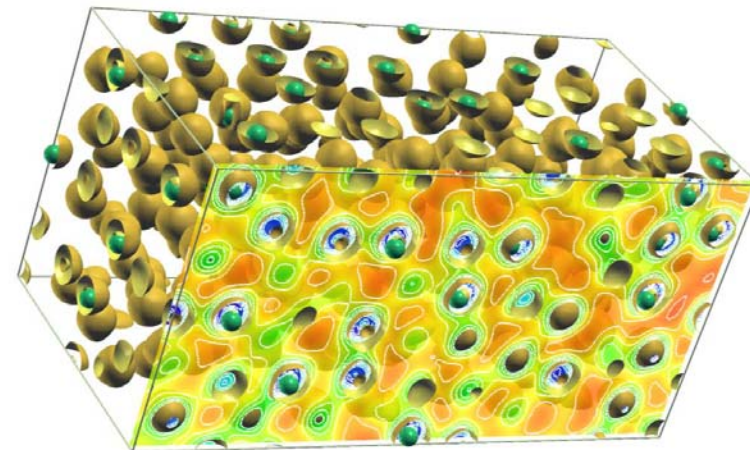
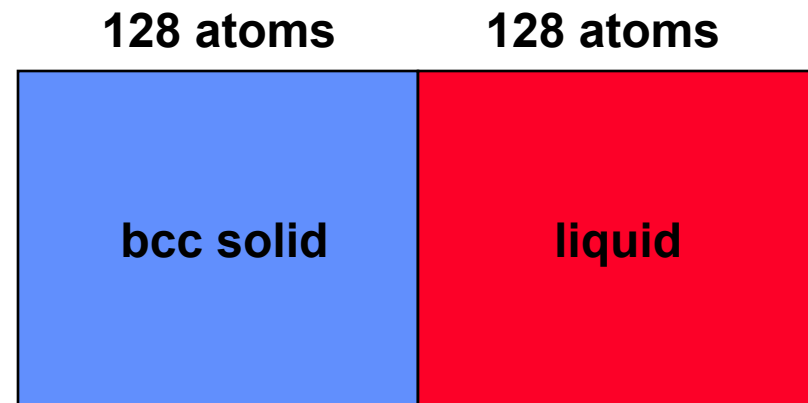
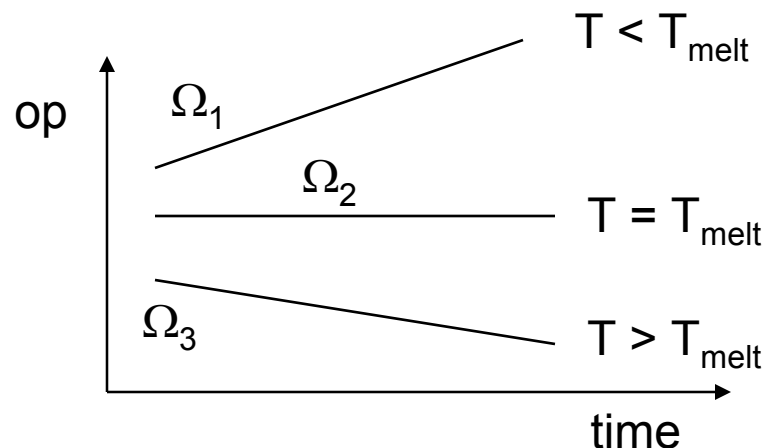


- **Parameters of simulation**

- $N = 256$  atoms (minimum size)
- constant  $T$ , constant  $N\Omega$
- 1-2 ps runs
- 1 k point

- **Diagnostic strategy**

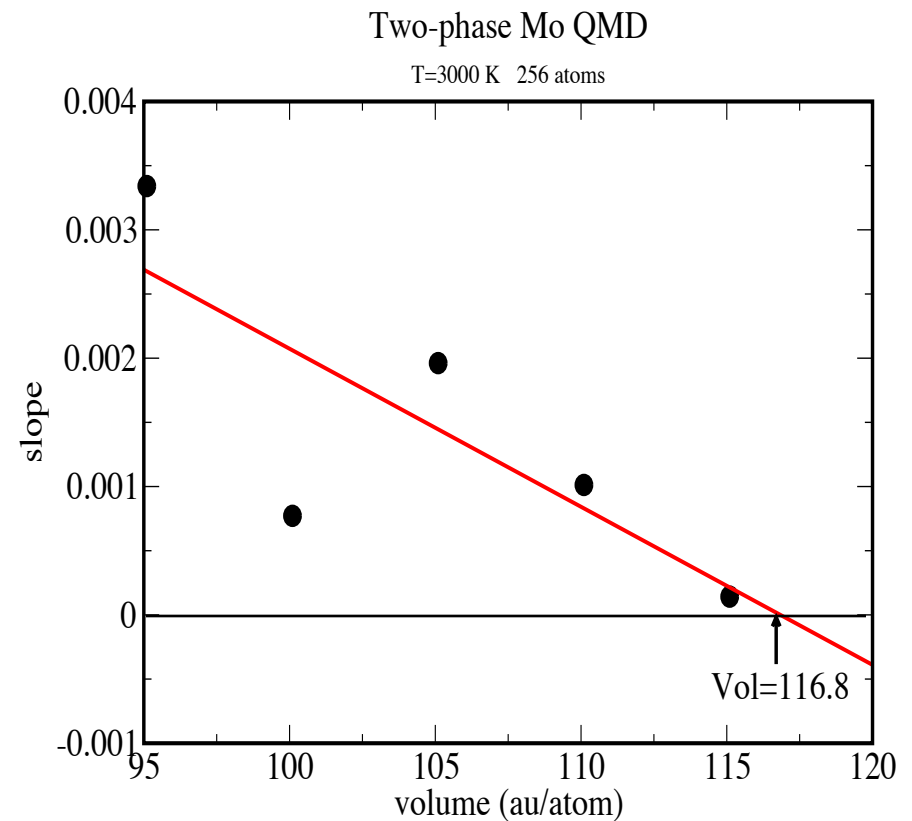
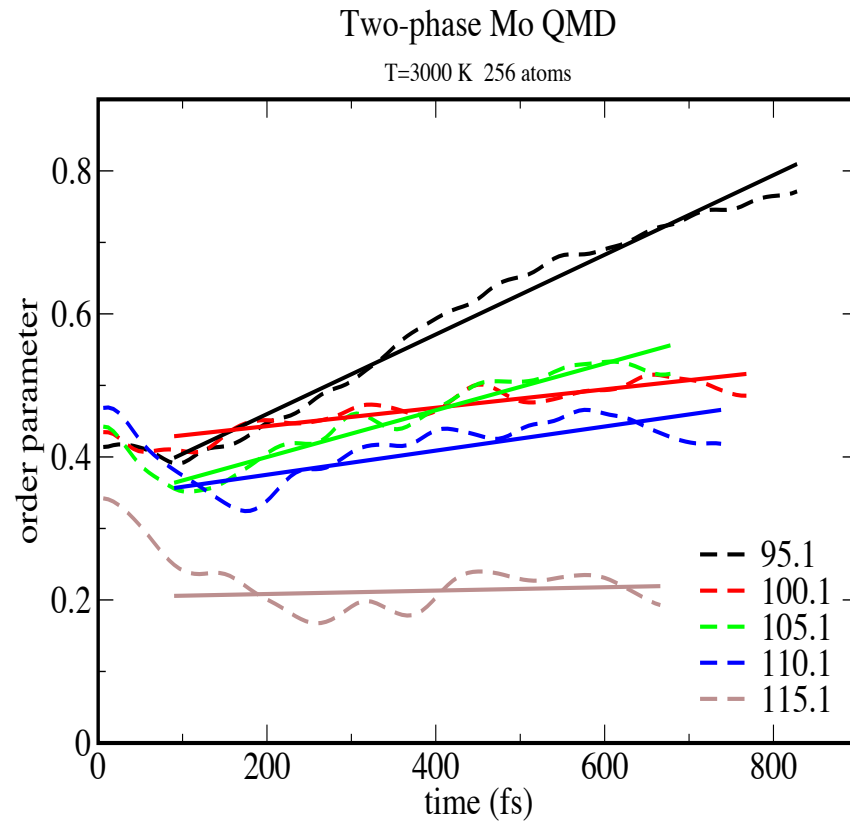
- monitor order parameter (op) rather than interface position
- for fixed  $T$ , do runs at several volumes of interest



**Snapshot of Mo QMD melt:**

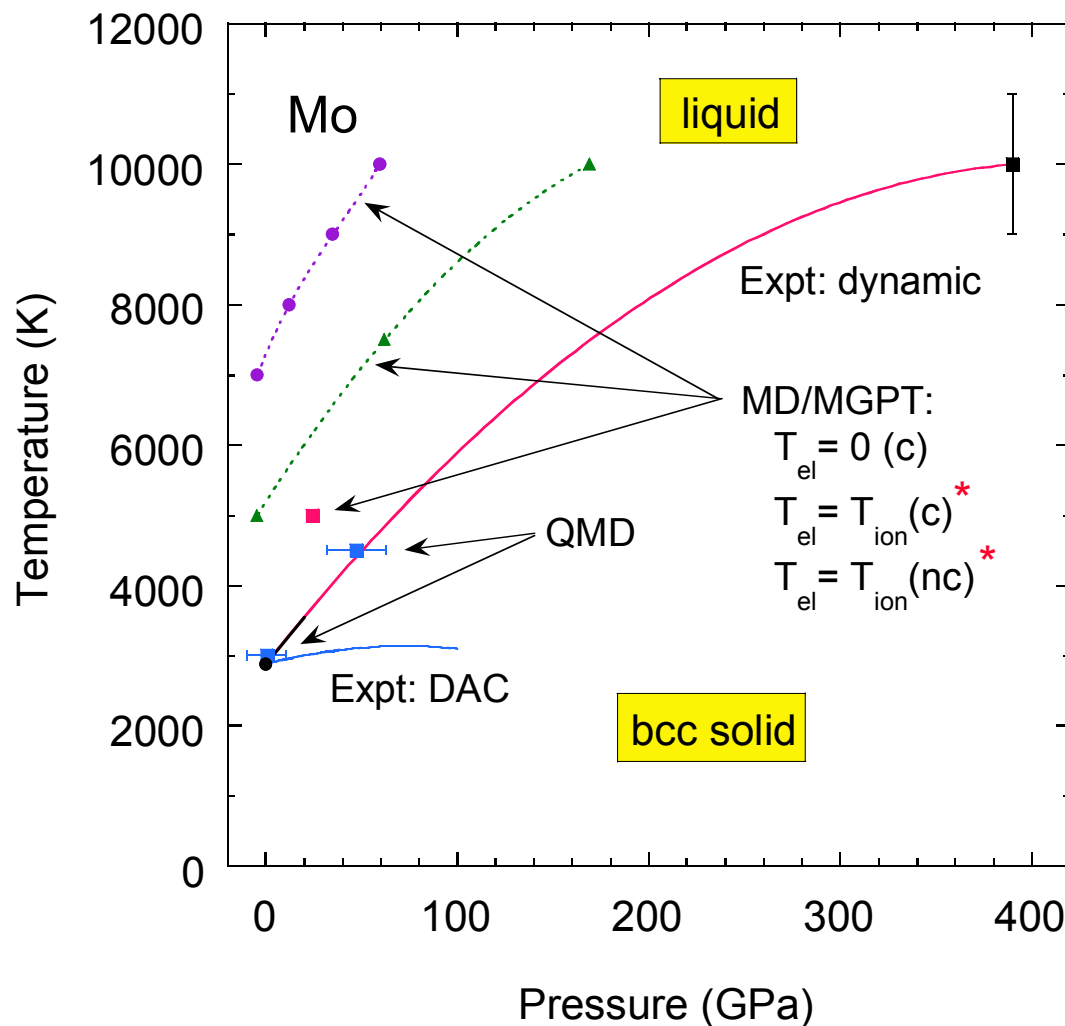
$T = 3500$  K and  $P = 35$  GPa

# Two-phase QMD melt analysis: an example



**Same methodology applies to MD/MGPT melt simulations**

# Mo melt curve: sensitive test of MGPT potentials



For MD/MGPT results:

(c) = *canonical* d bands

(nc) = *non-canonical* d bands

\* early forms of T-dependent MGPT potentials

**Additional simulated melt points from QMD and MD/MGPT with current potentials are in progress**



## Conclusions and expected scientific impact

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- **A new hybrid QMD-MD/MGPT simulation algorithm has been developed to obtain *quantum-level accurate* interatomic potentials**
  - excellent first-generation MGPT potentials have been developed in a Mo prototype
  - in the coming year, we expect to complete algorithm optimization, potential transferability testing, and initial large scale applications

**If we are successful ...**

- **New predictive science tool for materials extending continuously from *nanometers* to *micrometers***
  - essential to SciDAC goal of first-principles SCC multiscale modeling
  - permit fundamental investigations beyond current capabilities
- **Will enable accurate advanced programmatic applications**
  - non-equilibrium multiphase equations of state with phase kinetics
  - dynamic strength and failure modeling up to micron length scales